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PHYSICAL AND MECHANICAL PROPERTIES OF CAST 17-4 PH
STAINLESS STEEL*

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ABSTRACT

The physical and mechanical properties of an overaged 17-4 PH stainless steel casting have been examined. The tensile and compressive properties of cast 17-4 PH are only influenced to a slight degree by changing test temperature and strain rate. However, both the Charpy impact energy and dynamic fracture toughness exhibit a tough-to-brittle transition with decreasing temperature--this transition being related to a change in fracture mode from ductile, dimple to cleavage-like. Finally, although the overaged 17-4 PH casting had a relatively low room temperature Charpy impact energy when compared to wrought 17-4 PH, its fracture toughness was at least comparable to that of wrought 17-4 PH. This observation suggests that prior correlations between Charpy impact energies and fracture toughness, as derived from wrought materials, must be approached with caution when applied to cast alloys.

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INTRODUCTION

Prior studies of 17-4 PH stainless steel (1-11) have generally considered the mechanical and physical properties of wrought product forms, that is rolled plate, forgings, etc. There are, however, many instances where, because of economic considerations, 17-4 PH stainless steel castings might be an attractive alternative. Unfortunately, little information exists on the mechanical and physical properties of 17-4 PH stainless steel castings. This report presents the results of an evaluation of such a casting. Where available, direct comparison with data obtained from wrought 17-4 PH stainless steel is also included.

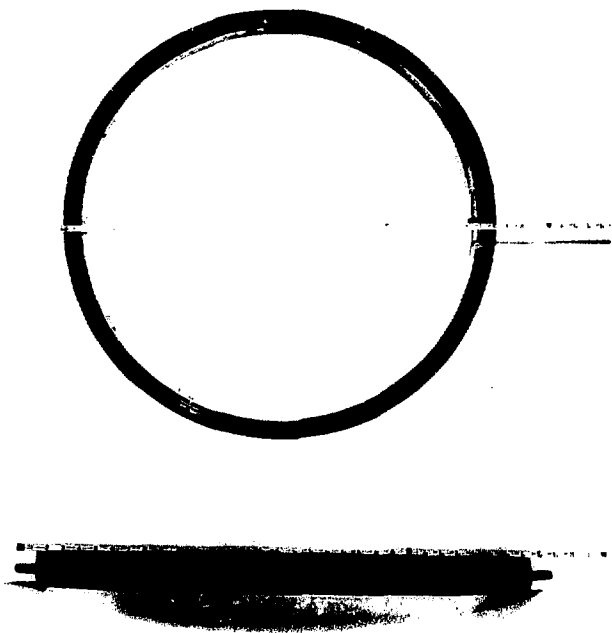


Figure 1. Top and Side Views of 17-4 PH Stainless Steel Seal Casting.

EXPERIMENTAL PROCEDURE

Figure 1 shows the 17-4 PH stainless steel casting evaluated in this study. This casting was selected since it is currently being considered as the primary metallic seal for a liquid metal breeder reactor spent fuel shipping container. As such, the seal must operate at temperatures between 298 and 473K. In addition, it must be able to withstand applied strain rates approaching 10 sec^{-1} .

Physical Properties

Physical property measurements of the 17-4 PH stainless steel castings involved determinations of the linear expansion, specific heat and thermal diffusivity as a function of temperature. A dual fused silica pushrod Theta dilaterometer¹ operating in a room temperature environment was used to obtain linear expansion measurements in the temperature range 298 to 1173K (12). Measurements between 298 to 2117K were made with a single fused silica pushrod dilaterometer, again methanol cooled in a room temperature environment. Finally, the linear expansion samples, 25.4 mm in length x 2.54 mm square, were equilibrated for one hour at each test temperature prior to expansion measurements.

Specific heat determinations utilized a Perkin Elmer Model 1050 differential scanning calorimeter connected to a Pk1 minicomputer-based digital data acquisition system (13). The thermal diffusivity results were obtained using a computer controlled laser flash diffusivity technique (14). From the specific heat, c_p , and the thermal diffusivity α , the thermal conductivity, k , was then calculated from

$$k = \rho c_p \alpha \quad (1)$$

where ρ is the density corrected for changes in temperature relative to room temperature (298K).

¹The dilaterometer was calibrated using standard fused silica and platinum samples.

Mechanical Behavior

The elastic properties of the 17-4 PH stainless steel castings were measured over the temperature range 233 to 1073F using standard ultrasonic techniques (14). These techniques require that the travel time, t , for an ultrasonic wave to propagate through a known specimen length, L , be obtained as a function of temperature. Once this travel time is known, the ultrasonic velocity, V , can be determined from

$$V = L/t$$

where V is the ultrasonic velocity for temperature T and L is the specimen length as described above. The elastic moduli were calculated from the Lamé constants, λ and μ , according to the following

$$E = \frac{3\lambda + 2\mu}{1 + \nu} \quad (1)$$

Eq. 2.1.1.4

$$\nu = \frac{\lambda}{2(\lambda + \mu)} \quad (2)$$

where ν is Poisson's ratio, the shear modulus, μ , and λ are the Lamé constants, respectively. Thus, the lateral contraction of a specimen under load to increasing-decreasing temperature about 298K.

The elastic behavior of the 17-4 PH stainless steel castings was measured using standard tensile, compression and Charpy impact test methods (the precracked) test procedures. Figure 2 illustrates the geometry of these samples as recommended for 17-4 PH stainless steel castings.^{*} Tensile and compression tests were performed between strain rates of 1.3×10^{-3} and 1.2 sec^{-1} over a temperature range 215 to 1458K.

^{*}The actual sample configurations are given in more detail in Appendix 2.

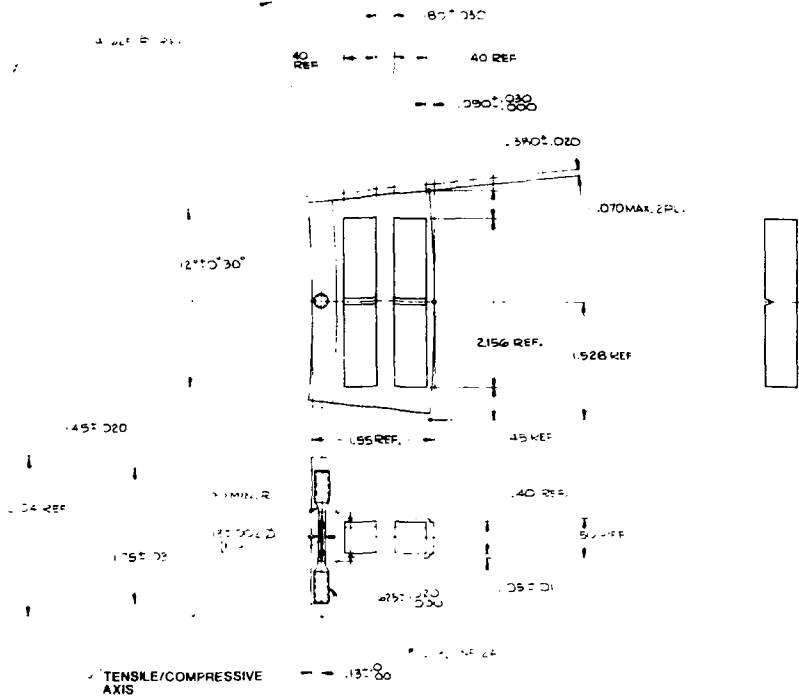


Figure 12-16. Detail view of the nut, for use in Figure 12-4. It has been enlarged by a factor of 2.54 cm.

The Charpy impact samples were tested in either the V-notched or fatigue precracked condition. Fatigue precracking utilized methods (5) where the final stress intensity during precracking, K_{fc} , was always controlled at less than one-half of the dynamic fracture toughness, K_{IC} . Both the notched and fatigue precracked samples were tested using an instrumented impact machine with the initial impact velocity being 3.0 m/s (11,13).

It was assumed that fracture was elastic; that is, no general yielding was observed, and the fracture toughness, K_{IC} , was obtained from the precracked samples by utilizing the relation:

$$K_{IC} = \frac{41.74}{B^{3/2}} \left[\left(\frac{W}{B} \right)^{1/2} + 4.1 \left(\frac{W}{B} \right)^{3/2} + 0.5 \left(\frac{W}{B} \right)^{5/2} + 0.1 \left(\frac{W}{B} \right)^{7/2} + 0.008 \left(\frac{W}{B} \right)^{9/2} \right] \quad (5)$$

where B is the thickness of the Charpy specimen, and W is the maximum deflection was observed, i.e., at the maximum temperature. The dynamic fracture toughness was obtained from a plot of K_{IC} versus the normalized value of the J integral, i.e.,

$$J_{11D} = (E_M/14)^{1/2} \quad (6)$$

where E_M is defined as

$$J_{11D} = 2E_M/BH \quad (7)$$

where B is the thickness and E_M was taken as the true specimen energy to maximum load (14).

RESULTS AND DISCUSSION

General

The chemical composition of the 17-4 PH stainless steel casting examined in this study is given in Table 1. Before machining, this casting had been homogenized at 1422K and then solution treated at 1011K. Final aging involved a four hour exposure at 922K. Optical microscopy indicated that the casting possessed an aged α -martensite matrix with δ -ferrite stringers, Figure 3. High magnification examination of the α -martensite matrix, Figure 4, indicated that the casting was in the overaged heat treatment condition; that it contained a rather coarse dispersion of the primary strengthening phase, spherical face centered cubic Cu particles. Further examination, Figure 5, revealed the presence of rod-shaped precipitates within the δ -ferrite stringers. X-ray energy dispersive analysis, Figure 6, indicated that these particles were relatively rich in Cu when compared to the α -ferrite matrix. The appearance of these rod-shaped Cu rich particles within the δ -ferrite stringers seems to be restricted to 17-4 PH stainless steel castings since their presence has not been reported in previous studies of wrought 17-4 PH stainless steel (1-11).

Table 1
Chemical Composition of 17-4 PH Casting

<u>Element</u>	<u>Weight Percent</u>
Cr	16.94
Ni	4.0
Cu	3.0
Mn	0.5
C	0.044
S	0.022
Si	0.7
Nb	0.3
Fe	Bal.

Figure 3. Optical Micrograph of 17-4 PH Stainless Steel Casting. White Areas α -Ferrite Stringers; Darker Matrix Aged α' -Martensite. Original Magnification 100X.



Figure 4. Transmission Electron Micrograph of Aged α' -Martensite in 17-4 PH Stainless Steel Casting Containing Spherical Cu Precipitates. Original Magnification: 40,000X.

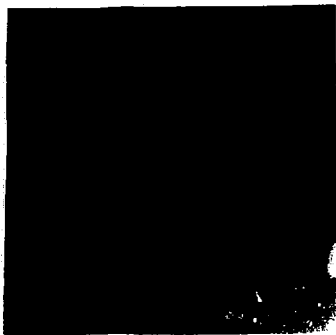
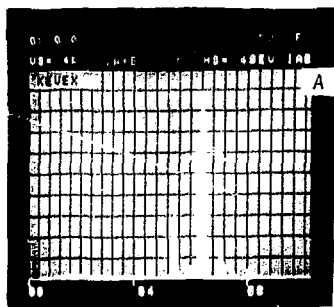
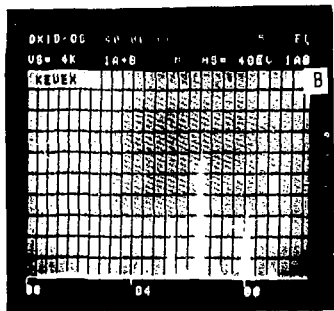


Figure 5. Transmission Electron Micrograph of δ -Ferrite Stringer Containing Rod-Shaped Cu-Rich Precipitate. Original Magnification: 52,000X.



Cr Fe Ni Cu



Cr Fe Ni Cu

Figure 6. X-Ray Energy Dispersive Spectra From (a) Ferrite Matrix and (b) Rod-Shaped Particles Shown in Fig. 5.

Fig. 11 is plotted.

The results of the thermal expansion measurements are tabulated in Table 2 and plotted in Figure 10. This figure shows that the thermal expansion of the 17-4 PH stainless steel is about 10% greater than that of the 304 stainless steel. The thermal expansion of the 17-4 PH stainless steel is about 10% greater than that of the 304 stainless steel. The thermal expansion of the 17-4 PH stainless steel is about 10% greater than that of the 304 stainless steel.

Fig. 11. Thermal expansion of 17-4 PH stainless steel.

Good agreement between the properties of 304 and 17-4 PH stainless steel was not determined for the thermal properties. The results of the thermal expansion and thermal conductivity measurements are tabulated in Table 3 and 4, respectively, with the calculated values of the thermal conductivity being presented in Table 5. This table for data is plotted in Figure 12 and 13, with the carbon lines in Figures 8 and 10 representing the values of previous data obtained from wrought 17-4 PH stainless steel (17). In general, these results indicate that the thermal properties of 17-4 PH stainless steel appear to be much more reasonable values in temperature than would be expected from the wrought 17-4 PH stainless steel data. Observations on the thermal properties of the 17-4 PH stainless steel indicate that the thermal conductivity of the two products may be due to the increased volume fraction of ferrite in each of the products, as well as the amount of

Table 2

Thermal Linear Expansion of 17-4 PH
Stainless Steel Castings

<u>Temperature (F)</u>	<u>$\Delta L/L_0$ (In.)</u>
279	-0.016
298	0.000
474	0.194
476	0.199
568	0.313
626	0.373
627	0.372
643	0.409
775	0.546
781	0.562
783	0.566
917	0.711
919	0.671
925	0.703
1064	0.824
1212	1.107

Table 3
Specific Heat of 17-4 PH Stainless Steel Casting

<u>Temperature (K)</u>	<u>Specific Heat (w sec gm⁻¹ °C⁻¹)</u>
350	0.4750
375	0.4884
400	0.4973
425	0.5054
450	0.5147
460	0.5220
475	0.5228
500	0.5335
525	0.5396
550	0.5477
575	0.5548
600	0.5636
625	0.5726
650	0.5805
675	0.5885
700	0.5978
725	0.6080
750	0.6341
775	0.6594
795	0.6812
800	0.6875
825	0.7240
850	0.7409
875	0.7576
900	0.7672
925	0.7780

Table 4

Thermal Diffusivity of 17-4 PH Stainless Steel Casting

<u>Temperature (Z)</u>	<u>Diffusivity (cm² sec⁻¹)</u>
294	0.0458
461	0.0452
627	0.0457
794	0.0440
961	0.0475
1127	0.0548

Table 5

Thermal Conductivity of 17-4 PH Stainless Steel Casting

<u>Temperature (K)</u>	<u>Conductivity (W cm⁻¹ K⁻¹)</u>
294	0.152
461	0.180
627	0.199
794	0.206
961	0.281

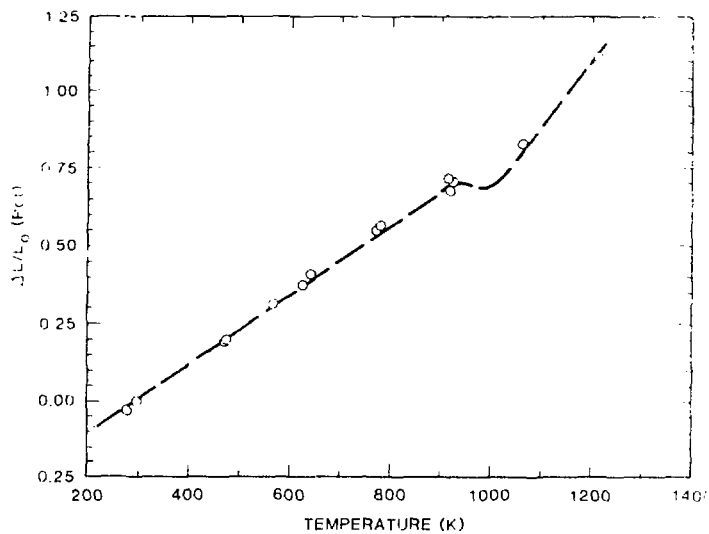


Fig. 10. Linear Thermal Expansion of 17-4 PH Stainless Steel. Data Points From 17-4 PH Castings, Dashed Line An Interpolation of Data From 17-4 PH Stainless Steel 1016-211.

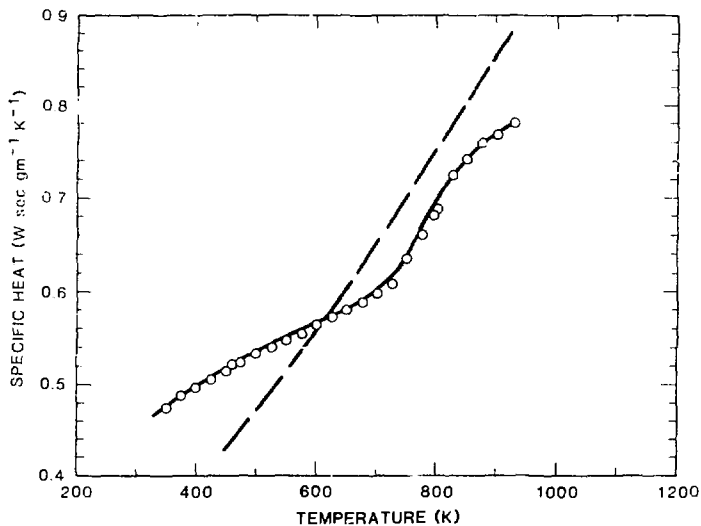


Figure 8. Specific Heat of 17-4 PH Stainless Steel as a Function of Temperature. Data Points from 17-4 PH Casting. Dashed Line an Average Obtained from Wrought 17-4 PH Stainless Steel (21).

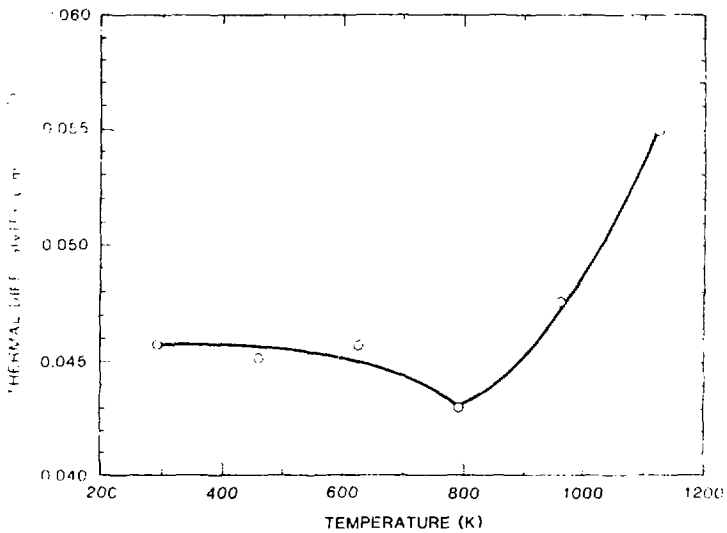


Fig. 6. Thermal Diffusivity of Cast 11-4 PH Stainless Steel as a Function of Temperature.

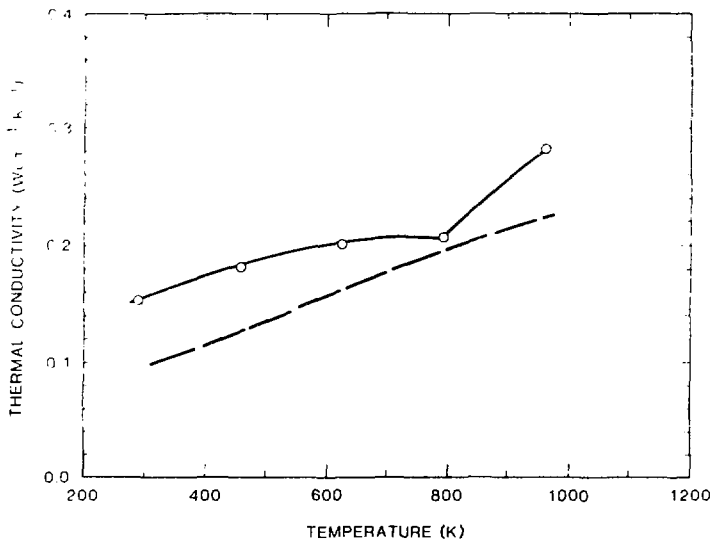


Figure 10. Thermal Conductivity of 17-4 PH Stainless Steel as a Function of Temperature. Data Points from 17-4 PH Castings, Dashed Line an Average of Values from Wrought 17-4 PH Stainless Steel. (2)

1315

1316 The test results for the 17-4 PH stainless steel

1317 specimens tested at temperatures of 400, 600, 800, and 1000

1318 degrees Fahrenheit are shown in Figure 1. The yield strength

1319 of the specimens increased from about 17-4 PH stainless steel

1320 yield strength of 120,000 psi at 400 degrees Fahrenheit to

1321 about 160,000 psi at 1000 degrees Fahrenheit. The tensile

1322 strength of the 17-4 PH stainless steel specimens tested

1323 at temperatures up to 800 degrees Fahrenheit showed a

1324 yield strength of about 140,000 psi and a tensile strength

1325 of about 170,000 psi. The yield strength of the specimens

1326 tested at 1000 degrees Fahrenheit was about 160,000 psi.

1327 The elongation of the specimens tested at 400, 600, and 800

1328 degrees Fahrenheit was about 12 percent. The elongation

1329 of the specimen tested at 1000 degrees Fahrenheit was

1330 about 14 percent.

1331 The test results for the 17-4 PH stainless steel specimens

1332 tested at 1200 and 1400 degrees Fahrenheit are shown in

1333 Figure 2. The yield strength of the specimens tested at

1334 1200 degrees Fahrenheit was about 140,000 psi and the

1335 tensile strength was about 170,000 psi. The yield strength

1336 of the specimens tested at 1400 degrees Fahrenheit was

1337 about 160,000 psi and the tensile strength was about

1338 190,000 psi. The elongation of the specimens tested at

1339 1200 degrees Fahrenheit was about 12 percent and the

1340 elongation of the specimens tested at 1400 degrees

1341 Fahrenheit was about 14 percent. The test results for

1342 the 17-4 PH stainless steel specimens tested at 1600

1343 and 1800 degrees Fahrenheit are shown in Figure 3.

1344 The yield strength of the specimens tested at 1600

1345 degrees Fahrenheit was about 180,000 psi and the tensile

1346 strength was about 220,000 psi. The elongation of the

1347 specimens tested at 1600 degrees Fahrenheit was about

1348 12 percent and the elongation of the specimens tested

1349 at 1800 degrees Fahrenheit was about 14 percent.

Table 6
 Young's Modulus and Poisson's Ratio
 of 17-4 PH Casting

<u>Temperature (K)</u>	<u>Young's Modulus (GPa)</u>	<u>Poisson's Ratio</u>
248	211.0	0.283
297	204.2	0.291
298	204.1	0.291
301	202.8	0.288
494	194.6	0.295
501	191.5	0.296
580	186.7	0.296
582	186.2	0.294
650	182.2	0.306
650	181.9	0.304
728	176.3	0.316
742	174.0	0.307
798	167.8	0.309
817	164.7	0.321
885	153.3	0.322
957	142.3	0.332
1031	134.0	0.344
1067	128.8	0.348
1151	118.0	0.359
1162	117.3	0.361

Table 1

Number of visits to the 174 fish distribution

<u>Year</u>	<u>Number of Visits</u>	<u>Number of Fish</u>
249	82,116	
248	79,207	
247	71,113	
242	68,130	
445	64,111	
507	61,111	
568	60,111	
680	51,111	
671	47,111	
675	41,111	
676	41,111	
747	40,111	
748	40,111	
749	40,111	
750	40,111	
751	40,111	
752	40,111	
753	40,111	
754	40,111	
755	40,111	
756	40,111	
757	40,111	
758	40,111	
759	40,111	
760	40,111	
761	40,111	
762	40,111	
763	40,111	
764	40,111	
765	40,111	
766	40,111	
767	40,111	
768	40,111	
769	40,111	
770	40,111	
771	40,111	
772	40,111	
773	40,111	
774	40,111	
775	40,111	
776	40,111	
777	40,111	
778	40,111	
779	40,111	
780	40,111	
781	40,111	
782	40,111	
783	40,111	
784	40,111	
785	40,111	
786	40,111	
787	40,111	
788	40,111	
789	40,111	
790	40,111	
791	40,111	
792	40,111	
793	40,111	
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795	40,111	
796	40,111	
797	40,111	
798	40,111	
799	40,111	
800	40,111	

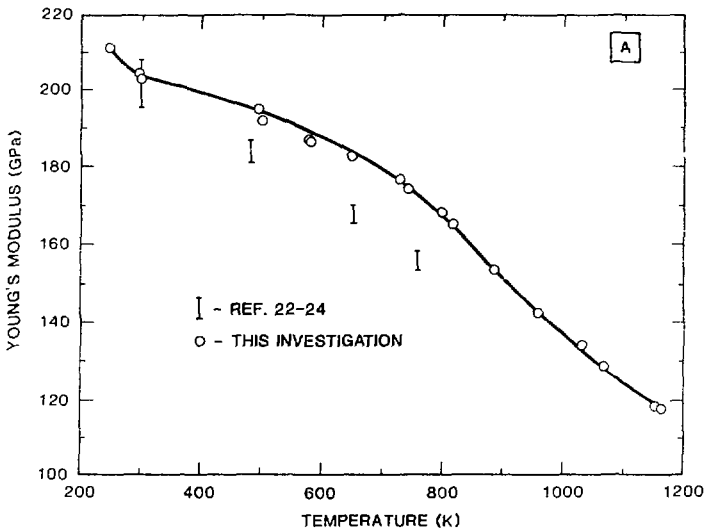


Figure 11. Elastic Properties of Overaged 17-4 PH Stainless Steel Casting (a) Young's Modulus, (b) Shear Modulus and (c) Poisson's Ratio.

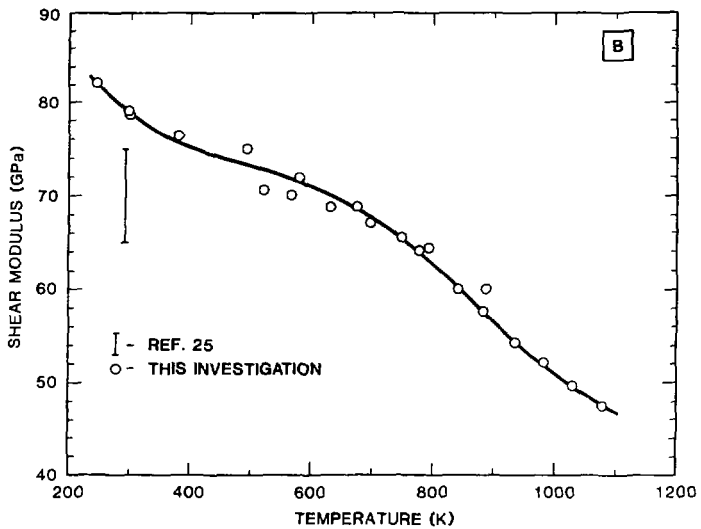


Figure 11. (Cont'd)

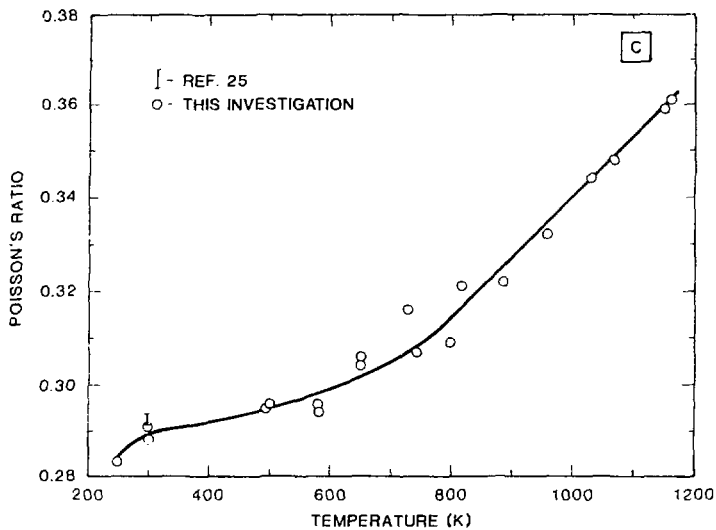


Figure 11. (Cont'd)

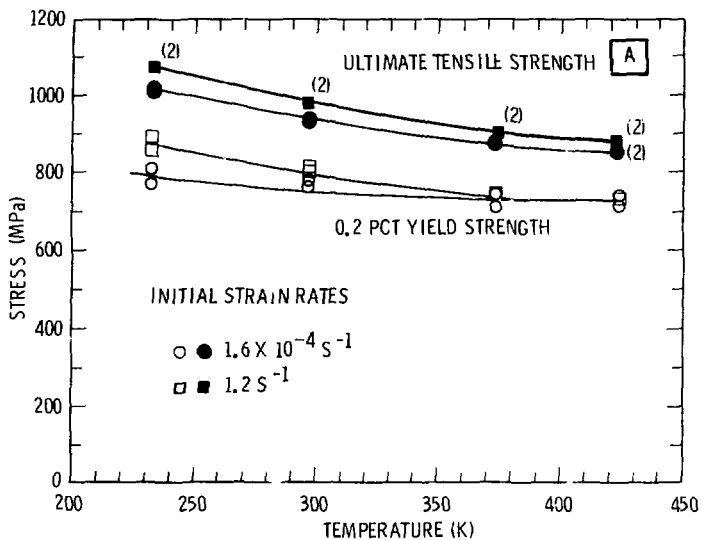
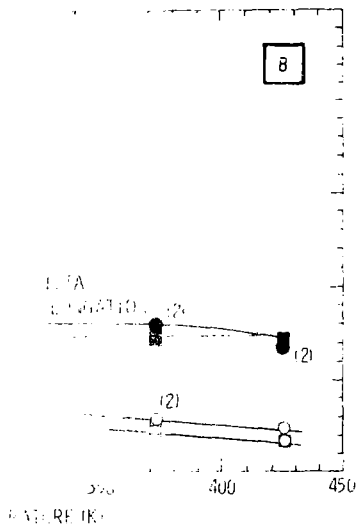


Figure 12. Influence of Test Temperature and Strain Rate on the Tensile Properties of Overaged Cast 17-4 PH Stainless Steel.



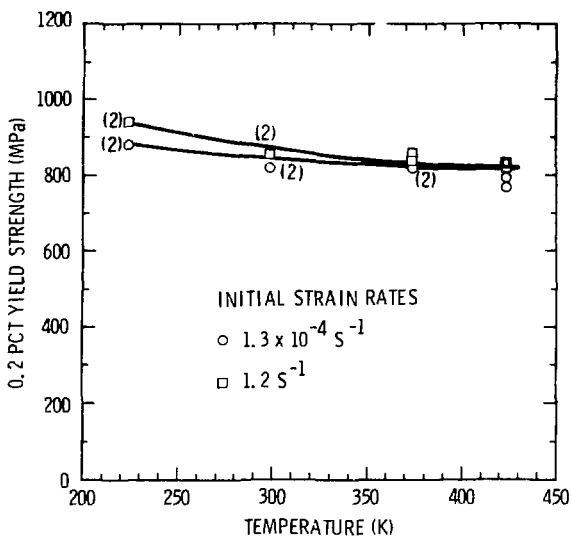


Figure 13. Influence of Test Temperature and Strain Rate on Compressive Yield Strength of Overaged Cast 17-4 PH Stainless Steel.

temperature and strain rate. Figure 12(b) shows that the uniform elongation decreased with both increasing test temperature and strain rate. This figure further indicates that, except at the lowest strain rate of $1.6 \times 10^{-4} \text{ s}^{-1}$, and the highest test temperature, 433K, the uniform elongation was independent of test temperature and decreased with increasing strain rate. Finally, fractographic examination showed that the tensile failure mode was, in all cases, characterized by the presence of transgranular dimples, with the larger dimples being associated with various inclusions and δ -ferrite, Figure 14.

Traditionally, the fracture toughness behavior of low strength steels and alloys has been examined by considering the influence of test temperature on the energy absorbed during impact fracture of a standard Charpy specimen. These investigations have typically shown that these steels undergo a tough-to-brittle transition with decreasing temperature, that is, there is a large reduction in absorbed energy within a relatively small temperature region. Figure 15 shows that the overaged 17-4 PH stainless steel casting under study also underwent such an energy related transition, but in both the values of the upper shelf energy and rate of energy decrease with decreasing temperature were less than those normally associated with lower strength alloys (26). If a typical 20 joule absorbed energy tough-to-brittle transition temperature criteria were applied to the overaged 17-4 PH, the T_{20J} transition temperature would have been approximately 350K, i.e., well above room temperature. Finally, comparison of the room temperature Charpy impact energy obtained for the overaged cast 17-4 PH ($E \sim 11$ joules) with that reported for wrought 17-4 PH tested at 866K (27) ($E \sim 37$ joules) suggests that cast 17-4 PH will absorb two-thirds less energy during impact loading than will wrought 17-4 PH.

Although the dynamic fracture toughness measurements--as shown in Figure 16--also exhibited such a tough-to-brittle transition behavior,

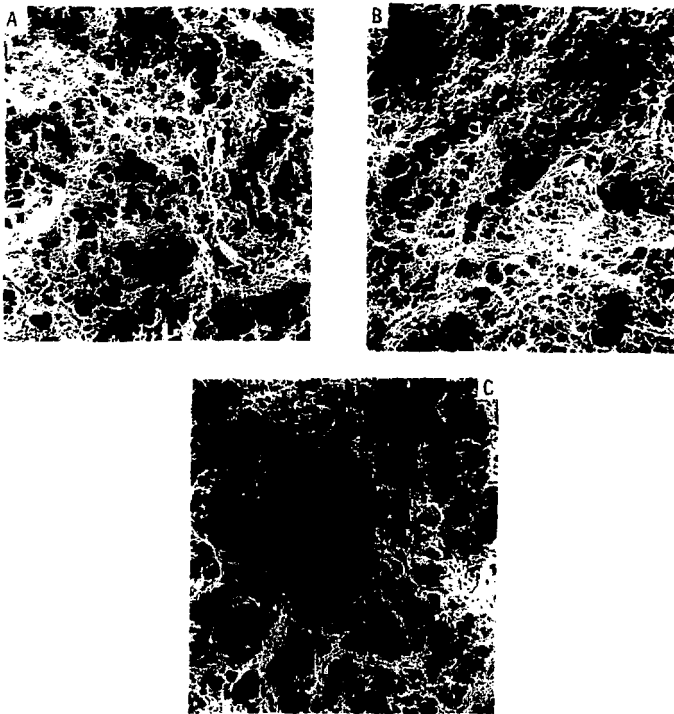


Figure 14. Scanning Electron Fractographs of Cast 17-4 PH Stainless Steel Tensile Samples Tested at:

- (a) $\dot{\epsilon} = 1.6 \times 10^{-4} \text{ s}^{-1}$, $T = 233\text{K}$;
- (b) $\dot{\epsilon} = 1.6 \times 10^{-4} \text{ s}^{-1}$, $T = 423\text{K}$; and
- (c) $\dot{\epsilon} = 1.2 \text{ s}^{-1}$, $T = 423\text{K}$.

Original Magnification: 400X.

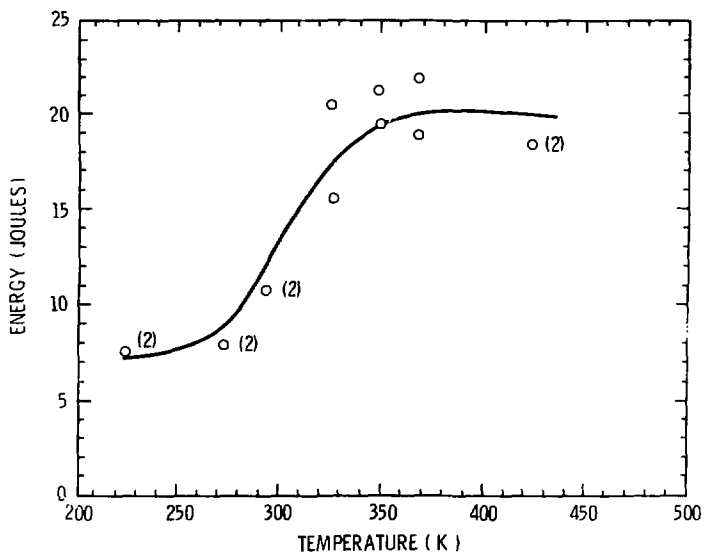


Figure 15. Charpy Impact Energy-Temperature Relationship in Cast 17-4 PH Stainless Steel.

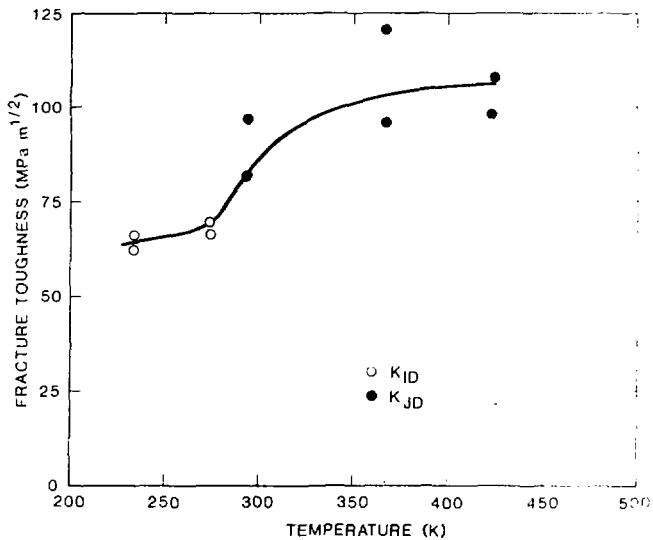


Fig. 16. Dynamic Fracture Toughness-Temperature Relationship for Cast 17-4 PH Stainless Steel.

the fracture toughness of the overaged 17-4 PH casting, even at the lowest test temperature examined, was still quite high, approximately $60 \text{ MPam}^{1/2}$. In addition, the room temperature toughness ($\sim 90 \text{ MPam}^{1/2}$) was at least comparable to that observed in wrought, overaged 17-4 PH (27), $K_{IC} \sim 130 \text{ MPam}^{1/2}$. These observations reinforce those of Floreen (28), wherein he concluded that Charpy impact energy-fracture toughness correlations previously suggested for wrought products are generally not applicable to castings, that is, the latter's Charpy impact values are typically quite low, even though their fracture toughness properties may be high.

Finally, fractographic examination of the Charpy V-notch and pre-cracked samples indicated that the fracture toughness transitions described above could be related to a change in fracture mode. At temperatures above 350K, failure in both types of samples involved microvoid initiation and growth, Figure 17(a). Decreasing the test temperature below 350K resulted in the introduction of increasing amounts of cleavage-like failure, Figure 17(b).

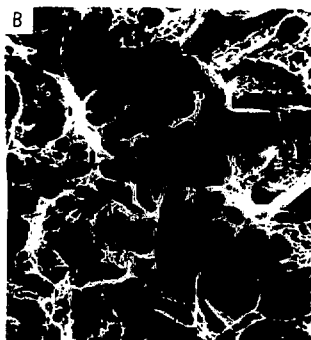
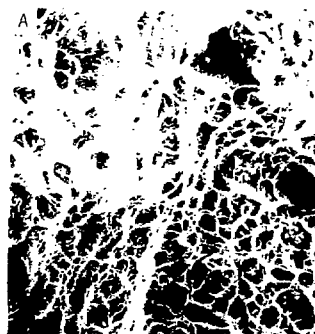


Figure 1

Figure 1 shows two histological sections of the choroid. Panel A shows a dense network of melanocytes and a prominent, thickened layer of Bruch's membrane. Panel B shows a similar structure, but with a more pronounced thickening of Bruch's membrane and a more dense network of melanocytes.

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SUMMARY AND CONCLUSIONS

This investigation has examined the physical and mechanical properties of an averaged 17-4 PH stainless steel casting and compared these properties, where available, with those for wrought 17-4 PH stainless steel. The study has shown that--

1. The linear expansion behavior of cast 17-4 PH is identical to that of the wrought alloy.
2. The thermal properties, specific heat, thermal diffusivity and thermal conductivity, of cast 17-4 PH stainless steel are more sensitive to temperature than is wrought 17-4 PH, that is, they vary in a more complicated fashion with temperature than do the thermal properties of wrought 17-4 PH.

The elastic properties, Young's modulus and the yield strength, tend to be higher in cast 17-4 PH stainless steel than in the wrought alloy, although they both decrease with increasing temperature.

The tensile and compressive properties of cast 17-4 PH were to some degree a function of test temperature and strain rate, although not to the same extent as for lower strength ferrous steels.

The Charpy V-notch impact energy and the dynamic fracture toughness both exhibited a temperature transition with decreasing test temperature. This transition was related to a change in fracture mode from ductile, dimple to brittle, cleavage.

The Charpy impact energy for cast 17-4 PH was generally less than that of wrought 17-4 PH, but the dynamic fracture toughness of cast and wrought 17-4 PH were comparable. This reinforces previous suggestions that Charpy impact-fracture toughness correlations obtained for wrought steels may not be applicable to castings.

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APPENDIX A

Mechanical Property Sample Configurations

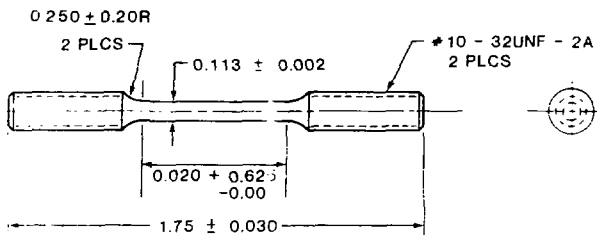


Figure A-1. Subsize Tensile Specimen.

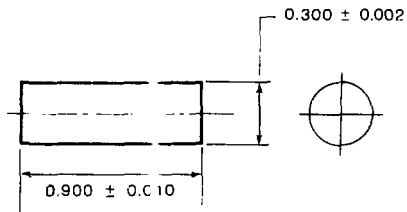


Figure A-2. Cylindrical Compression Specimen.

All dimensions in inches.

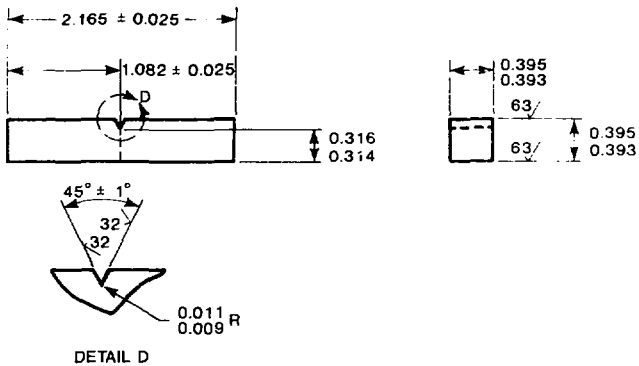
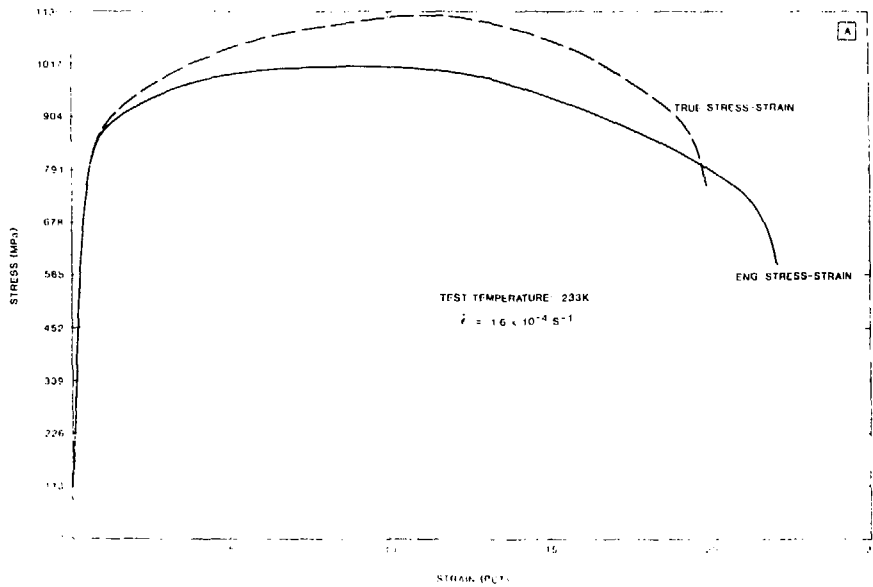


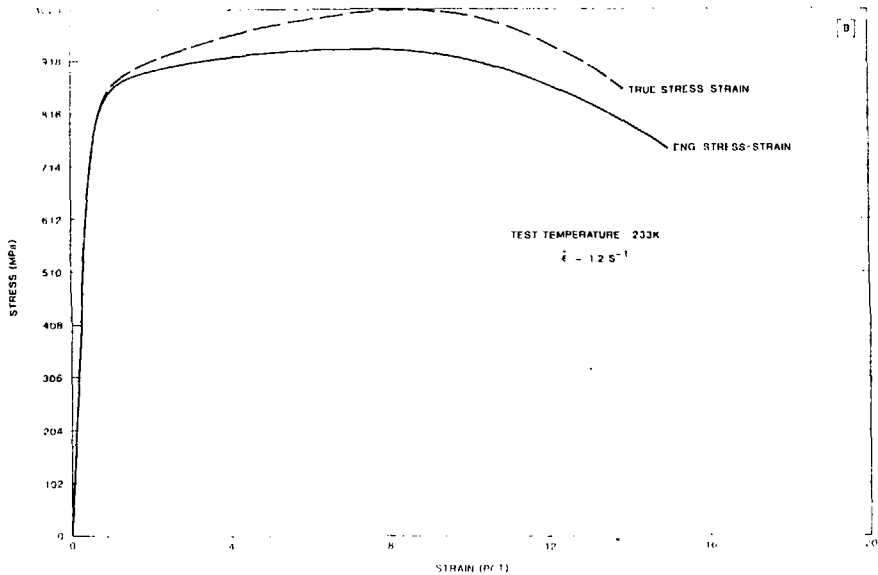
Figure A-3. ASTM Standard Charpy Impact Specimen.

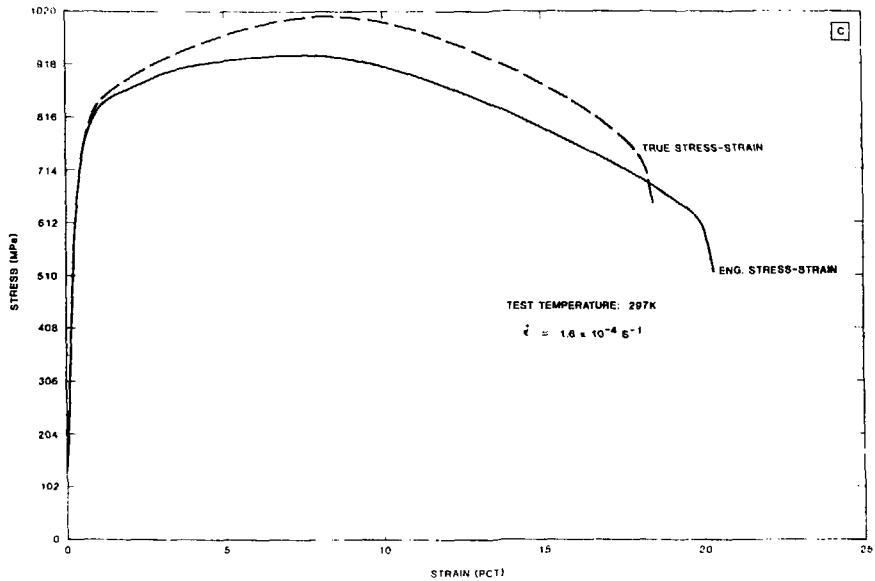
APPENDIX B

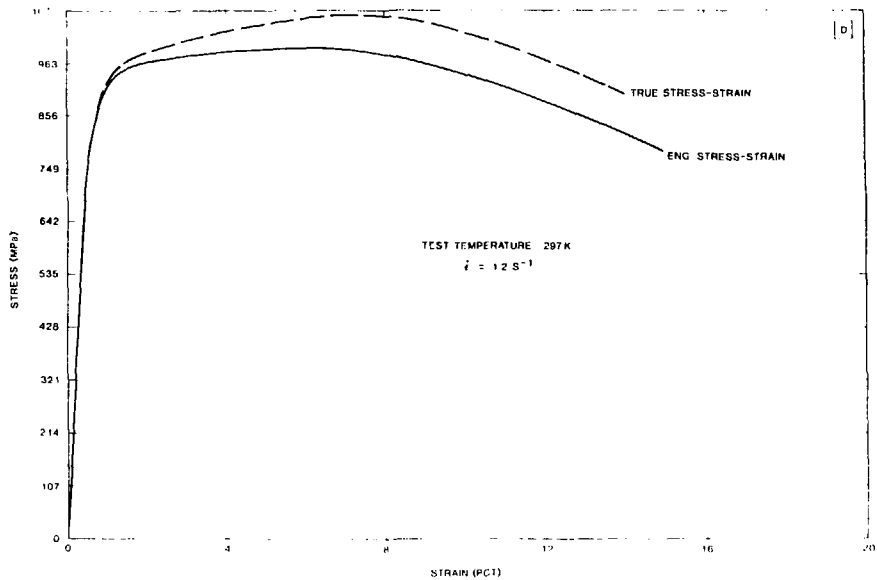
Representative Tensile Stress-Strain Curves for Overaged
17-4 PH Stainless Steel Casting

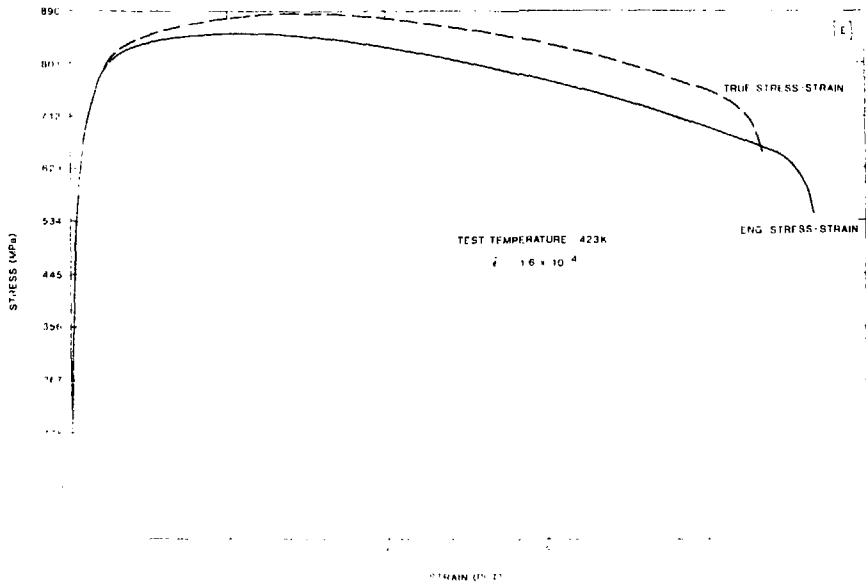
<u>Figure</u>	<u>Test Temperature (K)</u>	<u>Strain Rate (s^{-1})</u>
H-A	233	1.6×10^{-4}
E-B	233	1.2
H-C	297	1.6×10^{-4}
E-D	297	1.2
H-E	433	1.6×10^{-4}
E-F	433	1.2

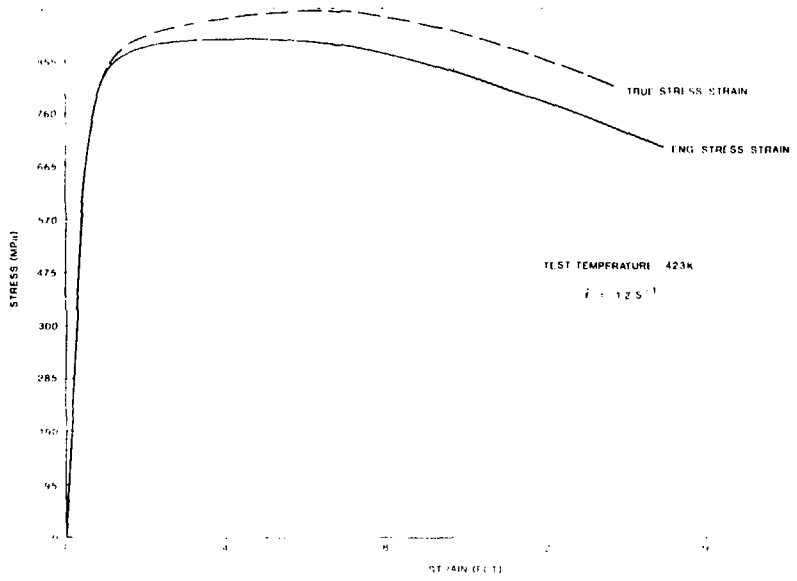












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APPENDIX

Representative Compressive Stress-Strain Curves
for American 1143 PH Stainless Steel, 1981

Figure	Test Temperature, °F.	Strain Rate, $\text{in./in.}^2/\text{min.}$
1-1	212	1.1×10^{-3}
1-2	212	1.0
1-3	212	1.1×10^{-3}
1-4	212	1.0
1-5	412	1.1×10^{-3}
1-6	412	1.0

